The Synthesis and Characterization of Cobalt-Rice Husk Silica Nanoparticles

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Abstract: Spherical and needle-like cobalt nanoparticles incorporated on rice husk silica was synthesized by a simple, rapid and eco-friendly sol-gel route at room temperature. This was achieved by direct mixing of 20 wt.% Co^{2+} with so-dium silicate from rice husk under ambient conditions to form RH-20Co. The cobalt was found to be isomorphously substituted in the silica matrix. A good dispersion of cobalt nanoparticles was evidenced from X-ray diffraction. Transmission electron microscopy confirmed the formation of well-dispersed and spherical cobalt nanoparticles with an average particle size of 5.6 nm. The magnetic properties of this material were measured with vibrating magnetometer. The saturation magnetization, M_s and coercive force, H_c was found to be 0.245 emu/g and 304.09 Oe respectively.

Keywords: Cobalt nanoparticles, spherical, rice husk silica, sol-gel, magnetization.

1. INTRODUCTION

Nanoparticles have been regarded as an exceptional and important material due to its well-established properties such as size and surface effects [1, 2]. These materials are applicable in miscellaneous fields such as catalysis, electronics, optical, biotechnology and magnetism [3-5]. Among known transition metals, cobalt undoubtedly has gained wide interest in numerous catalytic studies. For example, cobalt nanoparticles supported on silica in steam reforming ethanol [6], CoFe₂O₄ and Co₃O₄ nanocrystals as heterogeneous catalyst for cyclohexane oxidation [7] and also methane combustion [8]. Owing to its broad applications, various methods have been employed by researchers to produce unique Co or Co bimetallic nanoparticles which includes thermal decomposition [9, 10], pulsed laser irradiation [11], chemical vapor condensation (CVD) [12], co-precipitation [13], reversed micelle technique [14], reduction of cobalt chloride [15] and thermolysis [16]. Most recently, a ferromagnetic cobalt nanoparticle between 35-200 nm size range was successfully synthesized by a modified polyol technique [4]. Additionally, cobalt nanoparticles were synthesized by reverse microemulsion consisting of either neutral or ionic surfactants [17]. Furthermore, Fu and co-workers [18] have developed spherical core cobalt/silica nanoparticles by activated plasma evaporation of the metal. However, most of these existing procedures are expensive, complicated, time consuming and involves organic solvents. For instance, thermal decomposition methodology requires long period of calcination and high temperature as reported by Yang et al., [10].

Silica is an inert support material with a high surface area [19, 20]. It has been widely used in ceramics and electronics

[21]. Rice husk, an abundantly available agricultural waste has emerged as an ideal source of high grade amorphous silica. Many studies have been reported on the use of silica from rice husk as a catalyst support [22-25]. Sol-gel process is claimed to be a simple and effective synthesis pathway for the production of nanomagnetite [26], silica nanotubes [27] and nanocomposites [28, 29]. Moreover, recently, Zaky *et al.*, [30] obtained a narrow size distribution of silica nanoparticles with an average size of 50-70 nm from semiburned rice straw ash. Thus, herein, for the first time, we propose a simple and novel sol-gel route to synthesize cobalt nanoparticles supported on rice husk silica with unique morphology and magnetic properties.

2. MATERIALS AND METHODS

Rice husk (RH) was obtained from a local rice milling plant in Penang, Malaysia. Cobalt nitrate $(Co(NO_3)_2.6H_2O)$, R&M Chemicals, 99%) was used as the cobalt precursor. Sodium hydroxide (QRec, 99%) and nitric acid (QRec, 69%) were of analytical grade and used without further purification.

Washing and pretreatment of RH was carried out according to the procedure reported by Adam *et al.*, [24]. Washed and dried RH (30.0 g) was stirred in 500 mL of 1 M HNO₃ overnight, to enable the removal of metals from the RH [25]. This acid treated RH was washed thoroughly with copious amount of distilled water to ensure the absence of NO₃⁻ and subsequently dried in an oven at 110°C for 24 h. The resulting sample was denoted as RH-HNO₃. The weighed RH-HNO₃ was added to 500 mL of 1 M NaOH, stirred vigorously for 24 h, suction filtered to obtain the residue, RH-NaOH which was oven dried at 110°C for one day. RH-NaOH was weighed and recorded. The amount of silica extracted from RH was assumed to be the difference in the mass of RH-HNO₃ and RH-NaOH [25]. The filtered sodium

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silicate solution was kept in a covered container to be used as the source of silica. Preparation of RH-Silica was done following the published literature [22] but the pH was maintained at 3. RH-Silica was synthesized by titrating the sodium silicate with 3.0 M HNO₃ until it reached pH 3. A yellowish-brown suspension was observed during the titration. The observed suspension was aged for 2 days in a covered container. The gel was recovered by centrifugation and suction filtered, washed with copious amount of distilled water and oven dried at 110°C. The dried catalyst was ground to yield 3.30 g of RH-Silica.

Cobalt - rice husk silica was prepared by titrating the sodium silicate with 130 mL of 3 M HNO₃ containing 20 wt.% of $Co(NO_3)_2.6H_2O$ to obtain RH-20Co. Briefly, the titration was carried out until pH 3 and the suspension obtained was aged in a covered container at room temperature for 48 h. After 2 days of ageing, the resulting gel was washed with distilled water, centrifuged at 4000 rpm and finally recovered by suction filtration. The obtained solid material was subsequently oven dried at 110°C for 18 h and finely hand-ground to yield 3.25 g of RH-20Co.

The synthesized material was characterized by fourier transformed infrared measurements (FT-IR, Perkin Elmer System 2000), ²⁹Si MAS nuclear magnetic resonance technique (²⁹Si MAS NMR, Bruker DSX-3000), N₂ sorption analysis (Micromeritics Instrument Corporation model ASAP 2000, Norcross) and X-ray diffractometry (XRD, Kristallofex Siemens, model D5000) equipped with Cu K_a radiation ($\lambda = 1.50$ Å). The cobalt content was determined by atomic absorption spectroscopy (AAS, Perkin Elmer, model 3100). Prior to AAS analysis, the sample was digested with aqua regia and hydrofluoric acid solution. Material morphology was recorded on a transmission electron microscope (TEM, model Phillips CM12). Particle size measurements were conducted with Analysis Image Processing Cell A software. Mapping analysis was obtained by means of scanning electron microscope (SEM, model 50VP) equipped with Oxford Inca 400 Energy Dispersive X-ray (EDX). The morphology and roughness of the surface was determined by atomic force microscopy (AFM, ULTRAObjective) employing a non-contact mode. Magnetic property of the nanoparticles was investigated by vibrating sample magnetometer (VSM, DMS, model 8810) at 300 K, in the applied field sweeping from -10 to 10 kOe. The amount of sample used was 0.018g.

3. RESULTS AND DISCUSSION

3.1. Material Characterization

Fig. (1) illustrates the FT-IR spectra of RH-Silica and RH-20Co. The observed bands are similar to the work published elsewhere [31, 32]. The band at 967 cm⁻¹ in RH-Silica disappeared after introducing cobalt into the silica framework. This occurrence can be due to the presence of cobalt silicate or hydrosilicate [33]. Reaction between small cobalt particles with hydroxyl groups present on silica surface can form scarcely reducible cobalt silicate [14]. The inset spectrum in Fig. (1) clearly demonstrates a band at 660 cm^{-1} for RH-20Co which is absent in RH-Silica. This band is ascribed to spinel type Co-O bond corresponding to Co₃O₄ [34, 35]. It confirms the M-O stretching vibration in which Co²⁺ is tetrahedrally coordinated [36]. The presence of Co₃O₄ and CoO can be further evidenced from XRD results. This is in agreement with zirconia promoted silica supported cobalt catalyst subjected to air drying at 100°C which showed decomposition of cobalt nitrate into intermediate cobalt silicate phase and finally into Co_3O_4 [37].

The ²⁹Si MAS NMR spectra of the materials are shown in Fig. (2). RH-Silica exhibit Q^3 and Q^4 signals at 101.4 and 110.7 ppm. The Q^3 and Q^4 silicon centres are attributed to the (Si-O-)₃ Si(OH) and (Si-O)₄ Si groups respectively [38]. However, RH-20Co shows only an upward shift at -113.2 ppm, which corresponds to Q^4 signal with much broadening. Such broadening strongly suggests the interaction of silica with the paramagnetic cobalt which suggests the formation of Si-O-Co bond [39]. However, formation of cobalt nanoparticles (evidenced later) would have inhibited the



Fig. (1). The FT-IR spectra of RH-Silica and RH-20Co.



Fig. (2). The ²⁹Si MAS NMR spectra of RH-Silica and RH-20Co.

Table 1. The Textural Properties and Summary of Data Obtained from BJH and AAS Analysis

Catalyst	S_{BET} (m ² /g)	Average Pore Diameter (nm)	Average Pore Volume (cc/g)	^a Cobalt Loading (wt. %)
RH-Silica	312	3.0	0.27	-
RH-20Co	415	9.0	0.89	12.3

^aSample preparation via microwave-assisted acid digested system.

formation of Si-O-Co which may lead to the absence of Q^3 signal for RH-20Co.

The BET surface area (S_{BET}), pore volume and pore diameter obtained from N2- sorption and BJH analysis is shown in Table 1. Insertion of cobalt into the silica framework increased the SBET of RH-20Co ca. 33% in comparison to that of RH-Silica. It indicates a good dispersion of cobalt in the silica matrix [40]. This is in accordance with the XRD results (evidenced later). The pore diameter along with the pore volume of RH-20Co was also increased. Such characteristics can be attributed to the incorporation of larger cations such as Co²⁺ in a tetrahedral geometry of Si⁴⁺. It is indeed a strong indication that cobalt has been successfully incorporated into the silica matrix. Shi and Shen [41] found that Co-MCM-41 (5-20% Co) catalysts exhibited a similar increasing trend in pore diameter compared to that of Co-MCM-41. Enhancement in pore volume, pore diameter and surface area of RH-20Co may have favored the deposition of cobalt oxide species on the silica surface as suggested by Li and co-workers [42]. Chemical analysis by AAS for RH-20Co (Table 1) shows that ca. 12% of cobalt was successfully incorporated.

The N_2 adsorption-desorption isotherms together with the pore size distribution patterns of both materials are depicted in Fig. (3). Adsorption isotherms of RH-Silica and RH-20Co are of Type IV, indicative of mesoporous solids [43, 44]. However, RH-20Co exhibits a H1 hysteresis loop and RH-Silica shows H2 hysteresis loop suggesting a change in the pore structure with incorporation of cobalt. In addition, RH-Silica shows narrow and monomodal pore size distribution (PSD) with average diameter of 3.0 nm whereas RH-20Co exhibits a wider range of PSD with average pore diameter of 9.0 nm (Fig. **3b**).

Powder XRD pattern of the materials are presented in Fig. (4). A broad peak at $2\theta = 23^{\circ}$, shows the typical attributes of amorphous silica [45]. The profile of RH-20Co only shows weak and rather broad signals. Characteristic peaks of CoO, Co₃(OH)₄Si₂O₅ and Co₃O₄ can be identified as indicated in the pattern. This is in agreement with FTIR results discussed earlier (Fig. 1). A similar phenomenon was reported by Dutta *et al.*, [46] in the preparation of 10% Co/aerogel nanoneedles. Formation of CoO was most likely due to the reduction of Co₃O₄. Reflections corresponding to Co could not be observed which could be due to the overlap with other signals or broadening of the XRD peak. It could also be due to the presence of very small Co particles [47]. Smaller Co particles undetected in the XRD may also lead to untraceable crystalline peaks [48].

Selected TEM micrographs of RH-Silica and RH-20Co at different magnifications are presented in Fig. (5). Figs. (5a) and (b) demonstrate amorphous porous structure of RH-Silica. Highly-dispersed and spherical cobalt nanoparticles and needle-like nanoparticles can be seen in Figs. (5c) and (d) respectively. It is comparable with TEM results obtained for Co disulfide nanoparticles prepared by the reaction of



Fig. (3). The N_2 adsorption-desorption analysis of (a) the isotherms of RH-Silica and RH-20Co and (b) the pore size distribution of RH-Silica and RH-20Co.



Fig. (4). The XRD patterns of RH-Silica and RH-20Co.



Fig. (5). Representative TEM images of (a,b) RH-Silica and (c,d) RH-20Co at different magnifications and (e,f) SEM image and cobalt mapping analysis of RH-20Co.

CoCl₂.6H₂O with thiourea and hydrazine in aqueous solution at a refluxing temperature of 90-95°C [49]. In RH-20Co synthesized at pH 3, cobalt ions were fairly well dispersed without much agglomeration. This is presumably due to the pH range of 2-5 which favored adsorption of cobalt ions on negatively charged silica surface and thereby improve the cobalt dispersion [50]. Good dispersion of cobalt in RH-20Co could be further evidenced by SEM and cobalt mapping images as shown in Figs. (**5e**) and (**f**) respectively.

A narrow and uniform particle size distribution of RH-20Co with a mean diameter of 5.6 nm is shown in the histogram (Fig. 6). All the particles were observed as monomodal without formation of large particles. This shows that the simple sol-gel method used in this study is competent in producing cobalt particles with an average size of 5.6 nm. This is comparable to published methodology such as incipient wetness impregnation, drying and calcination in NO/He which produced 18 wt. % Co/SiO₂ nanoparticles of \pm 4.6 nm diameter [51].

AFM images of RH-20Co are displayed in Fig. (7). The surface topography of the material consists of spherical shaped pore system which is in line with TEM analysis. The surface roughness mean square (rms) of RH-20Co was found to be 107.24 nm.

3.2. Magnetic Properties

The magnetic hysteresis curve obtained at 300 K for RH-20Co is depicted in Fig. (8). The corresponding saturation magnetization (M_s) , coercivity (H_c) and remanent magnetization (M_r) were noted to be 0.245 emu/g, 340.09 Oe and 0.0115 emu/g respectively. M_s for RH-20Co was much lower compared to M_{bulkCo} = 166 emu/g [36]. Decrease in M_S is mostly due to the smaller cobalt rice husk silica nanoparticles synthesized in this study. Hard magnet behavior is shown from the hysteresis loop. Materials with large H_c (>100 Oe) are known as hard magnets [52]. The hysteresis of this material show the presence of a 'curvature' shape which indicates ferromagnetic (FM) nature and a 'straight' shape which correspond to antiferromagnetic (AFM) properties. Similar magnetic hysteresis plot was reported for CoO nanoparticles with size ranging from 10-80 nm prepared by sol-gel method [53]. Existence of CoO evidenced from FTIR and XRD results (Figs. 1 and 4) may lead to the antiferromagnetic attributes of RH-20Co. Co nanoparticles prepared in this work is most likely following the core-shell model, in which the core is attributed to metallic Co (ferromagnetic) and the shell consists of CoO spesies (antiferomagnetic) [54] as confirmed by TEM images in Fig. (9). The interfacial coupling between the core and shell



Fig. (6). Particle size distribution of the synthesized cobalt nanoparticles. Inset representative TEM image showing particles with average size 5.6 nm.



Fig. (7). The AFM images of RH-20Co.

system can lead to higher H_c and M_r and results in a lower M_s [55] as obtained in this study. However, the extent of the antiferromagnetic influence by residual Co phase is still under investigation.



Fig. (8). Magnetization (*M*) versus applied filed (*H*) curve of RH-20Co at 300K.



Fig. (9). TEM micrograph showing the presence of CoO shell for RH-20Co.

4. CONCLUSION

Cobalt rice husk silica nanoparticle was synthesized *via* a simple sol-gel method under mild conditions. This simple approach resulted in the successful generation of spherical and uniform cobalt nanoparticles ranging 2-15 nm. High BET surface area of RH-20Co suggests a good dispersion of cobalt in the siliceous matrix. The characterization by FT-IR and XRD reveals the presence of CoO, Co_3O_4 and Co silicate hydroxide in the material. The RH-20Co was found to exhibit both ferromagnetic and antiferromagnetic properties due to the core-shell system. Smaller size of the synthesized particles results in lower M_s value in comparison to M_s of bulk cobalt.

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